

*IN VIVO OXIDATION IN TKA  
COMPONENTS:  
A SPECTROSCOPIC AND  
NANOINDENTATION STUDY*

*Francisco J. Medel<sup>1</sup>, Hina Patel<sup>1</sup>, Alexis  
Cohen<sup>1</sup>, Steven M. Kurtz<sup>1,2</sup>*

*<sup>1</sup>Drexel University; <sup>2</sup>Exponent, Inc.  
Philadelphia, PA. USA*

# INTRODUCTION

- *In vivo* oxidation confirmed for traceable THA polyethylene components. This phenomenon may be more relevant in TKA.
- FTIR is commonly used to characterize oxidation in medical polyethylene.
- Raman spectroscopy and Nanoindentation provide microstructure and mechanical information, respectively.

# OBJECTIVE AND HYPOTHESIS

- **Global**
  - Compare the utility of FTIR, Raman spc., and nanoindentation to characterize mechanical and microstructure changes due to *in vivo* oxidation of historical TKA polyethylene tibial inserts.
- **Secondary**
  - These techniques would allow us to detect regional differences in the physical, chemical, and mechanical properties

# Knee Retrievals Information (n = 8)

- Processing route&resin:
  - Molded 1900H/Extruded GUR 415
- Implant designs:
  - Miller-Gallante I and II, Insall-Burstein II, AGC
- **Gamma-air sterilization**
- **Average Shelf Life: 0.6 years (0.2-1.0 y)**
- **Average Implantation Time: 11.5 years (8.3-13.0 y)**

## Clinical information

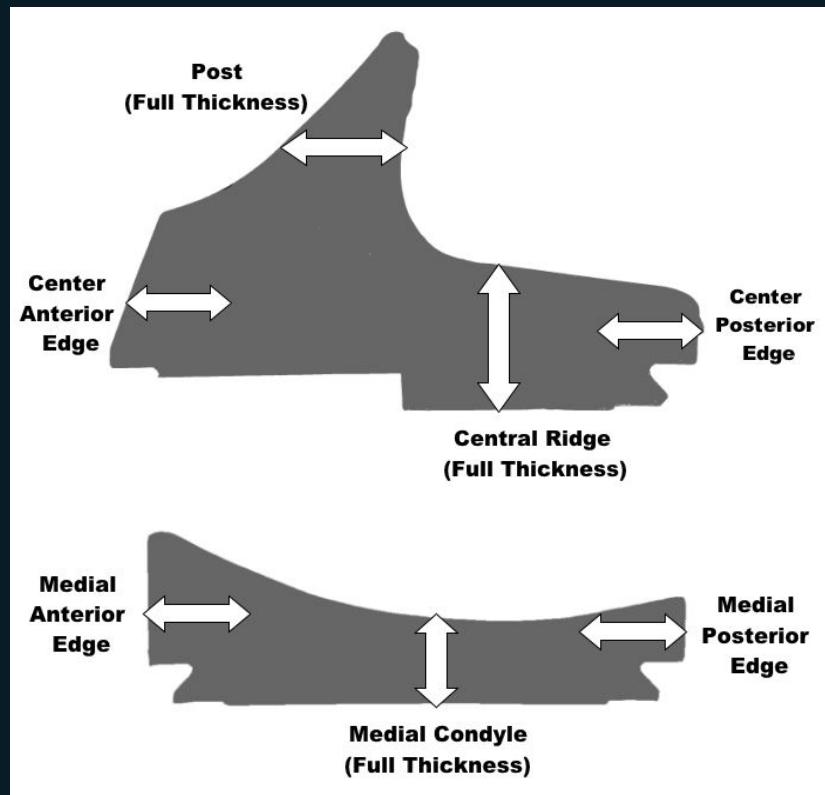
- Gender: 6F/2M
- Age at revision: 54-84 years
- Diagnosis at revision:
  - Loosening, PE Wear, Instability, Failed Patella, Metallosis

# FTIR spectroscopy

- Method

- 200 mm thick sections:
  - Medial Condyle
  - Unloaded central spine
- Boiled in heptane for 6 hours
- Scanned at 0.1 mm increments
- **Max OI** (ASTM F2102-01)
- **Max TVI** (ASTM F2381)

$$-\%C = \frac{\frac{A_{1897}}{A_{1303}}}{\frac{A_{1897}}{A_{1303}} + 1}$$

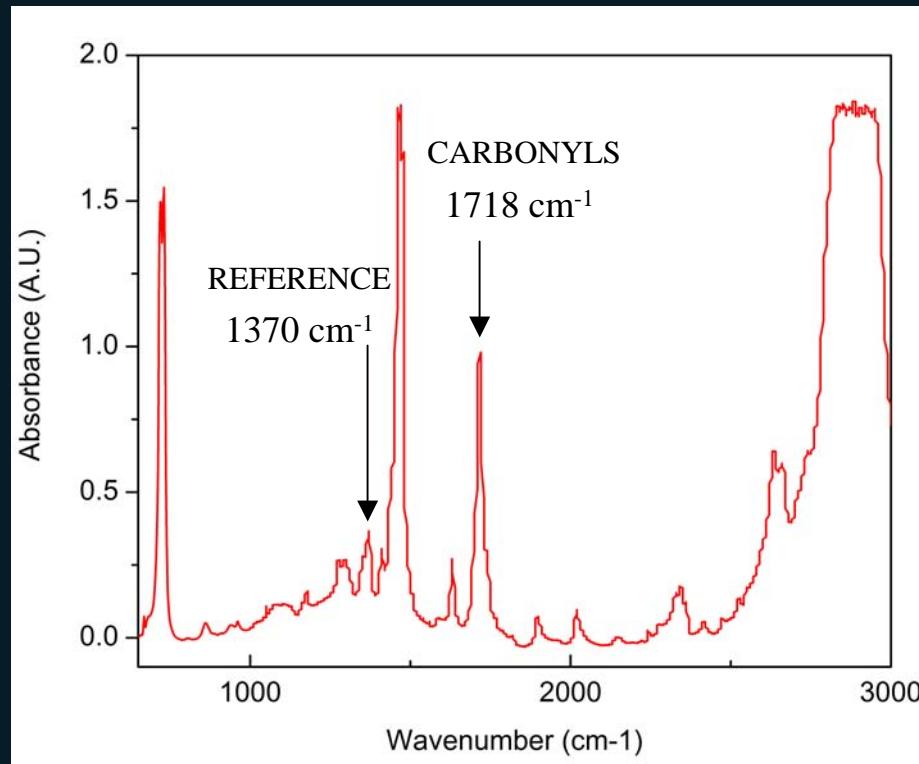


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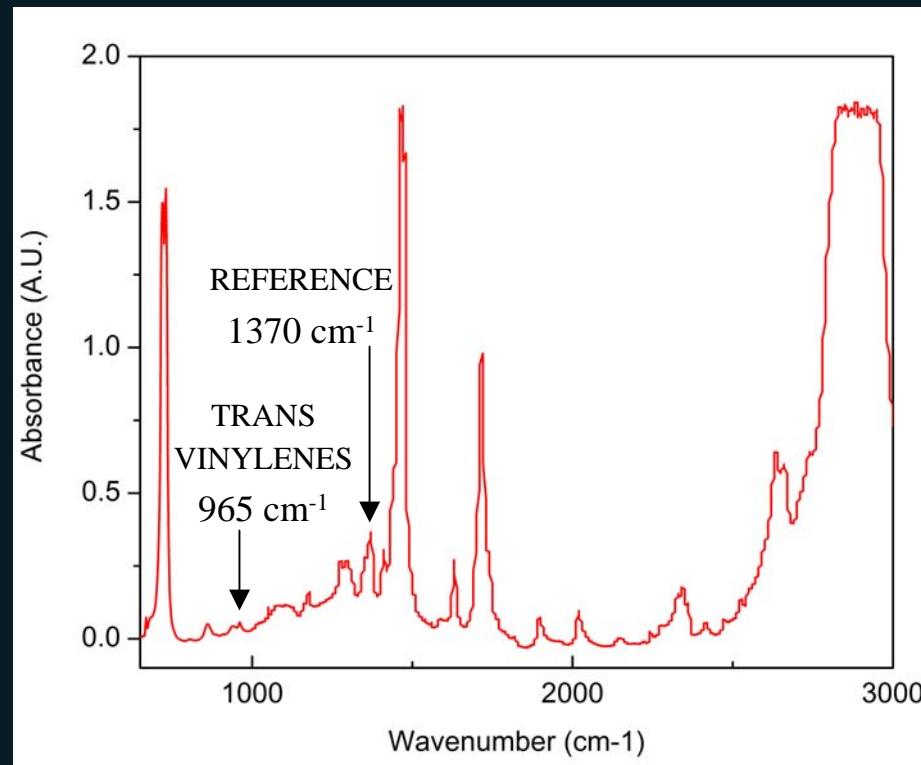


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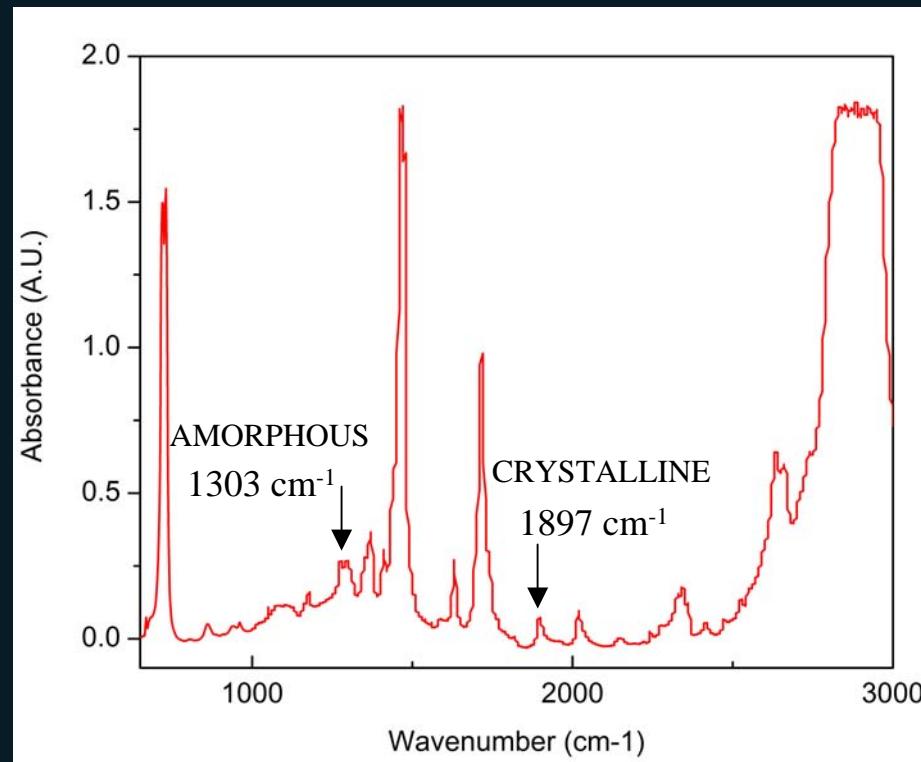


# FTIR spectroscopy

- Method

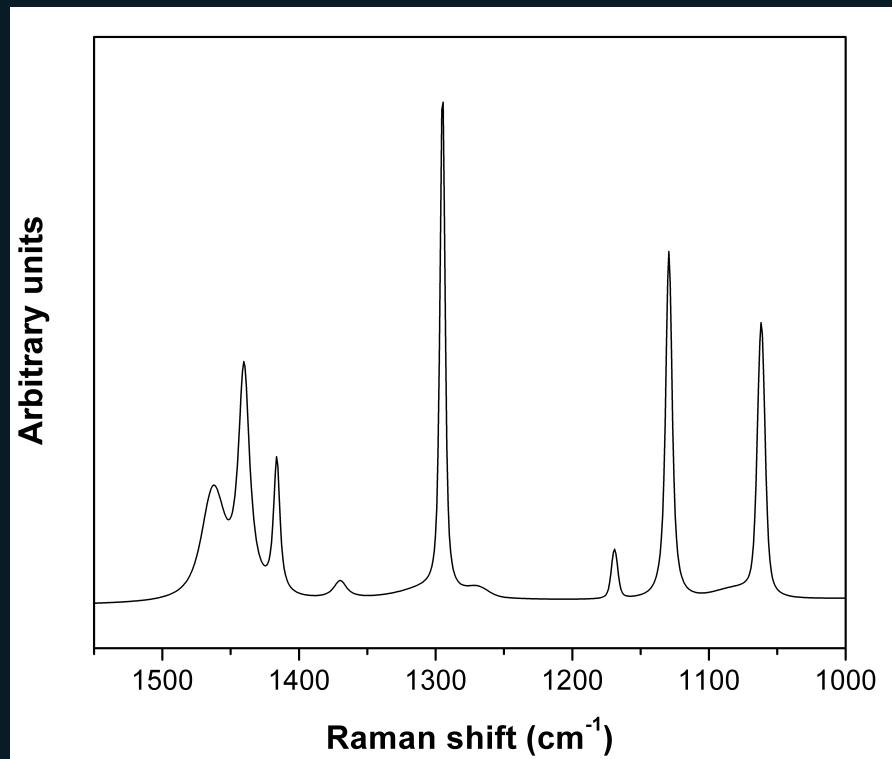
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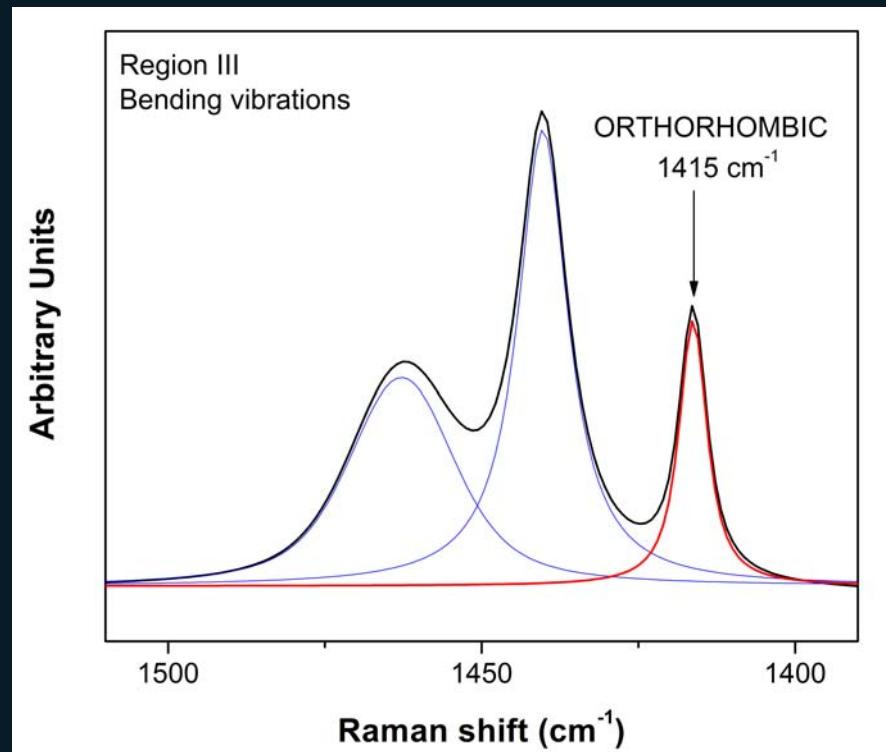
# Raman spectroscopy

- Method
  - Regions probed:
    - Surface, Subsurface and Bulk
  - Green laser line (514 nm)  
Initial power 25 mW
  - $1800 \text{ mm}^{-1}$  gratings.  
 $2\text{cm}^{-1}$  spectral resolution
  - Integration time ( $\sim 420$  s)
  - Properties measured
    - Orthorhombic Crystallinity
    - Amorphous fraction
    - Intermediate fraction
    - Overall Crystallinity



# Raman spectroscopy

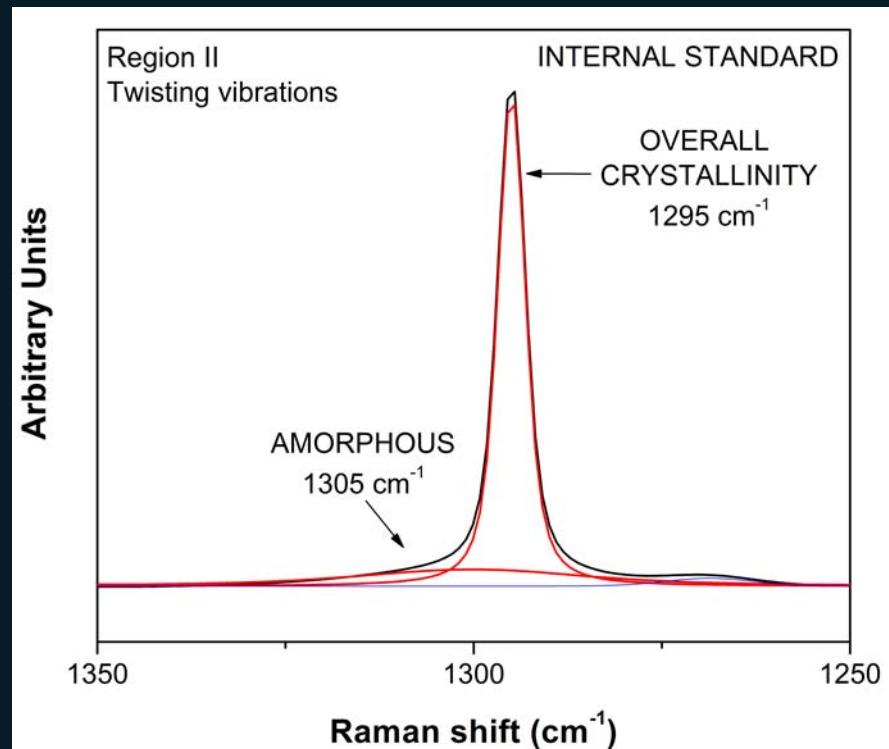
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$$\alpha_c = \frac{I_{1415}}{(I_{1295+1305+1269}) \times 0.45}$$

# Raman spectroscopy

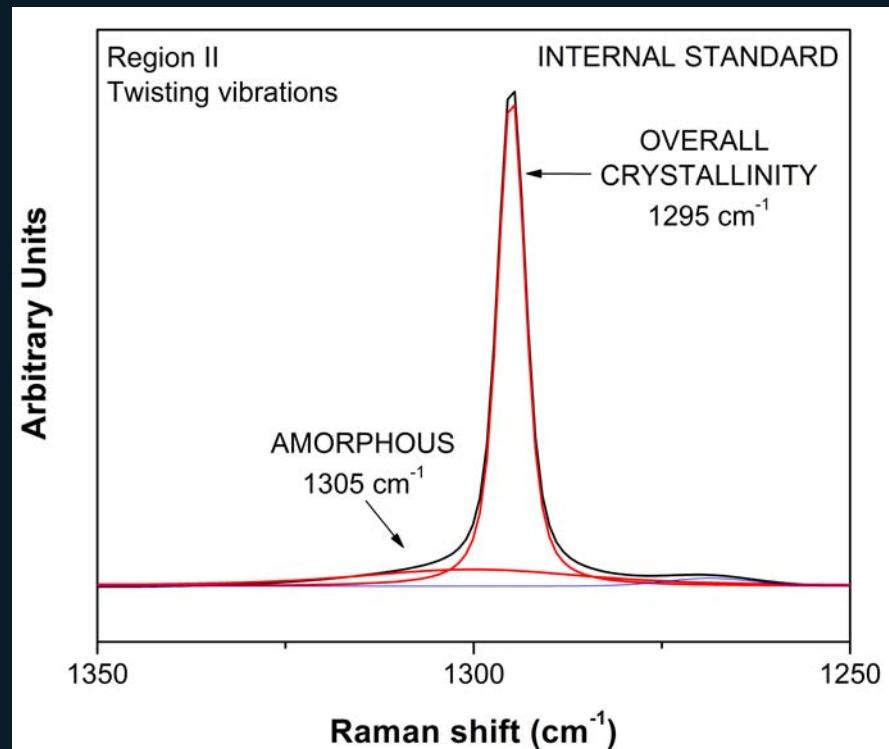
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  - Green laser line (514 nm)  
Initial power 25 mW
  - $1800 \text{ mm}^{-1}$  gratings.  
 $2\text{cm}^{-1}$  spectral resolution
  - Integration time ( $\sim 420$  s)
  - Properties measured
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    - **Amorphous fraction**
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    - Overall Crystallinity



$$\alpha_a = \frac{I_{1305}}{(I_{1295+1305+1269})}$$

# Raman spectroscopy

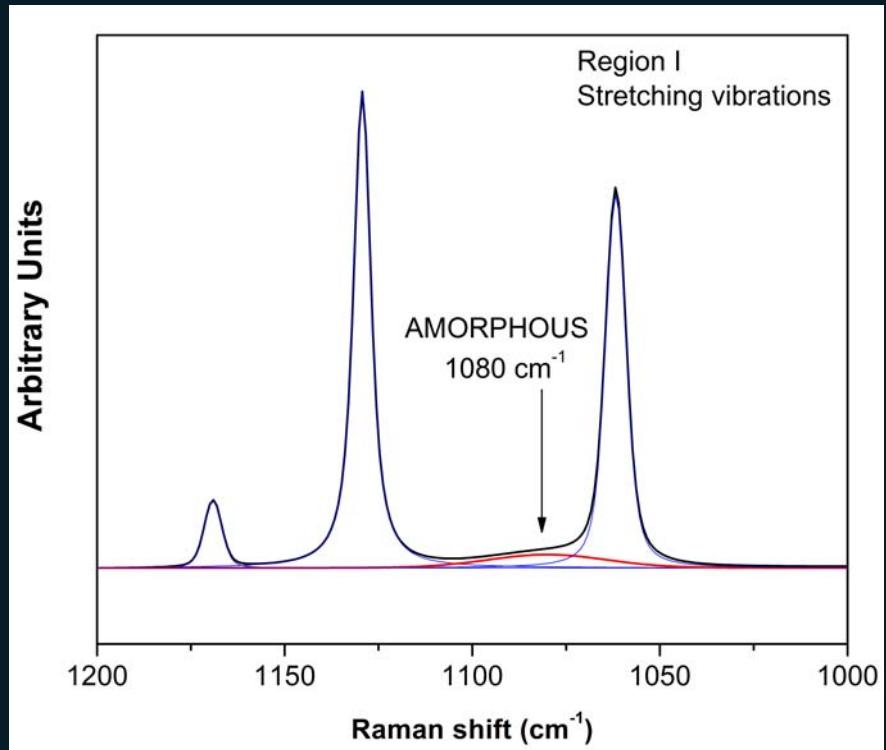
- Method
  - Regions probed:
    - Surface, Subsurface and Bulk
  - Green laser line (514 nm)  
Initial power 25 mW
  - $1800 \text{ mm}^{-1}$  gratings.  
 $2\text{cm}^{-1}$  spectral resolution
  - Integration time ( $\sim 420$  s)
  - Properties measured
    - Orthorhombic Crystallinity
    - **Amorphous fraction**
    - Intermediate fraction
    - Overall Crystallinity



$$\alpha_d = \frac{I_{1295}}{(I_{1295+1305+1269})}$$

# Raman spectroscopy

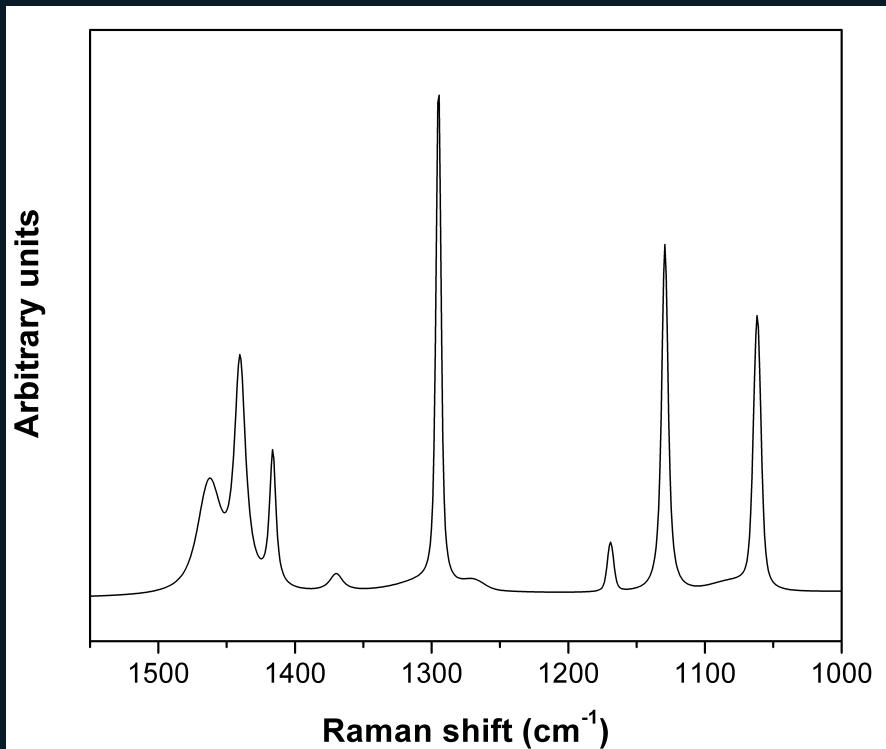
- Method
  - Regions probed:
    - Surface, Subsurface and Bulk
  - Green laser line (514 nm)  
Initial power 25 mW
  - $1800 \text{ mm}^{-1}$  gratings.  
 $2\text{cm}^{-1}$  spectral resolution
  - Integration time ( $\sim 420$  s)
  - Properties measured
    - Orthorhombic Crystallinity
    - **Amorphous fraction**
    - Intermediate fraction
    - Overall Crystallinity



$$\alpha_a = \frac{I_{1080}}{(I_{1295+1305+1269}) \times 0.80}$$

# Raman spectroscopy

- Method
  - Regions probed:
    - Surface, Subsurface and Bulk
  - Green laser line (514 nm)  
Initial power 25 mW
  - $1800 \text{ mm}^{-1}$  gratings.  
 $2\text{cm}^{-1}$  spectral resolution
  - Integration time ( $\sim 420$  s)
  - Properties measured
    - Orthorhombic Crystallinity
    - Amorphous fraction
    - **Intermediate fraction**
    - Overall Crystallinity



$$\alpha_b = 1 - (\alpha_c + \alpha_a)$$

# Nanoindentation

- Method

- Hemispherical tip ( $\phi 13.5 \mu\text{m}$ )

- Indentation depth  $4.5 \mu\text{m}$

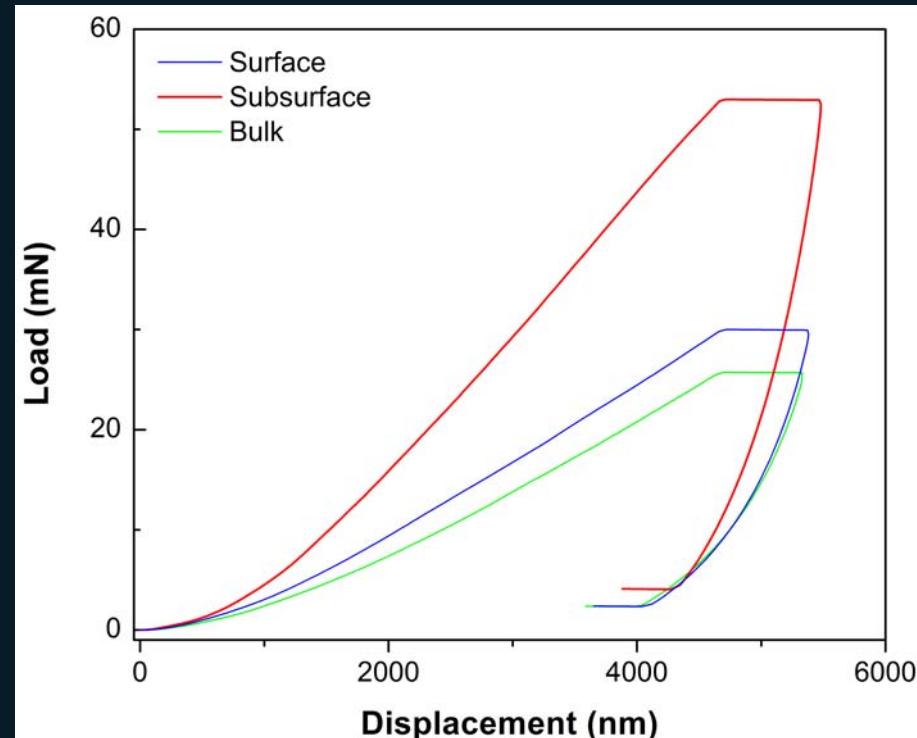
- 0.2 mm steps

- Hardness**

- (Oliver and Pharr method)**

- Elastic Modulus**

- (Sneddon equation)**



$$H = \frac{L_{\max}}{A} = \frac{L_{\max}}{\pi \left( \frac{h_{\max} + h_r}{2} \right) \left[ 2R - \left( \frac{h_{\max} + h_r}{2} \right) \right]}$$

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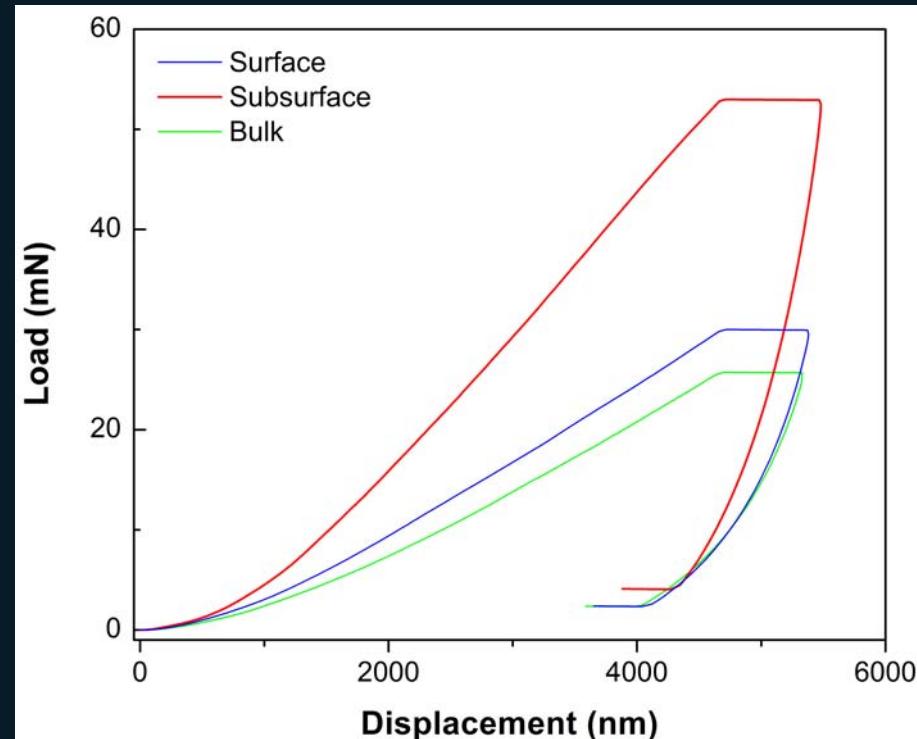
- 0.2 mm steps

- Hardness**

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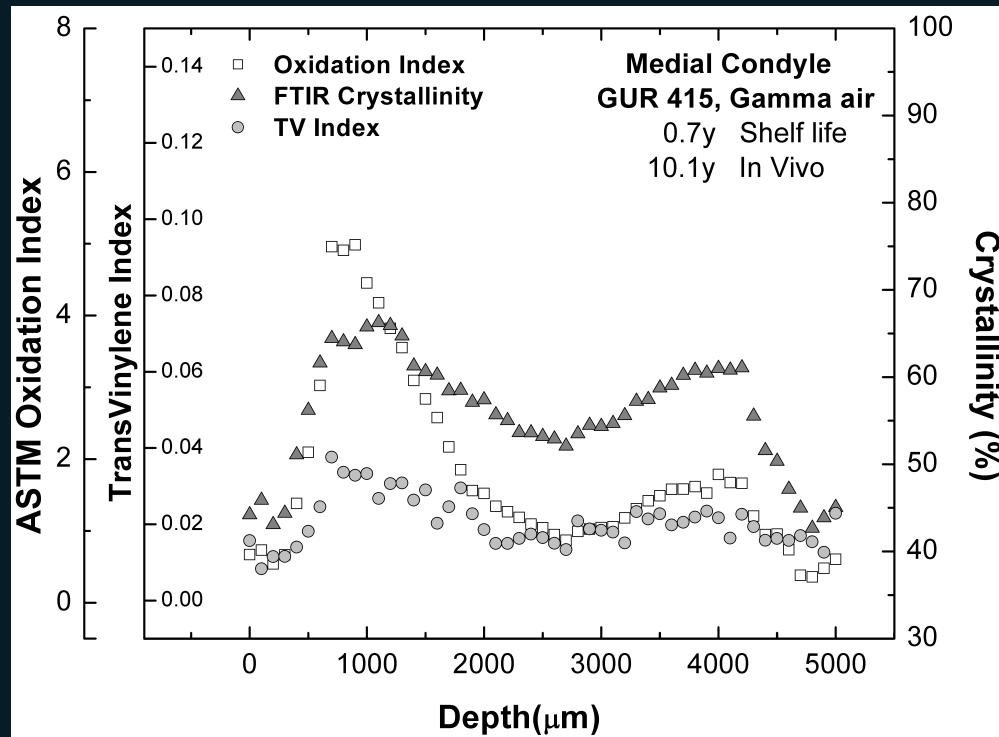


$$E = \frac{(1 - \nu^2)\sqrt{\pi}}{2\sqrt{A}} \frac{dl}{dx} = \frac{(1 - \nu^2)\sqrt{\pi}}{2\sqrt{A}} \frac{2L_{\max}}{h_{\max} - h_r}$$

# RESULTS

## OI, TVI and %C

- Property profiles showed **subsurface maxima** (~ 1 mm)
- **Backside less degraded** than the superior surface ( $p \leq 0.03^*$ )
- **Antero-posterior faces more degraded** than bearing surfaces ( $p < 0.05^*$ )



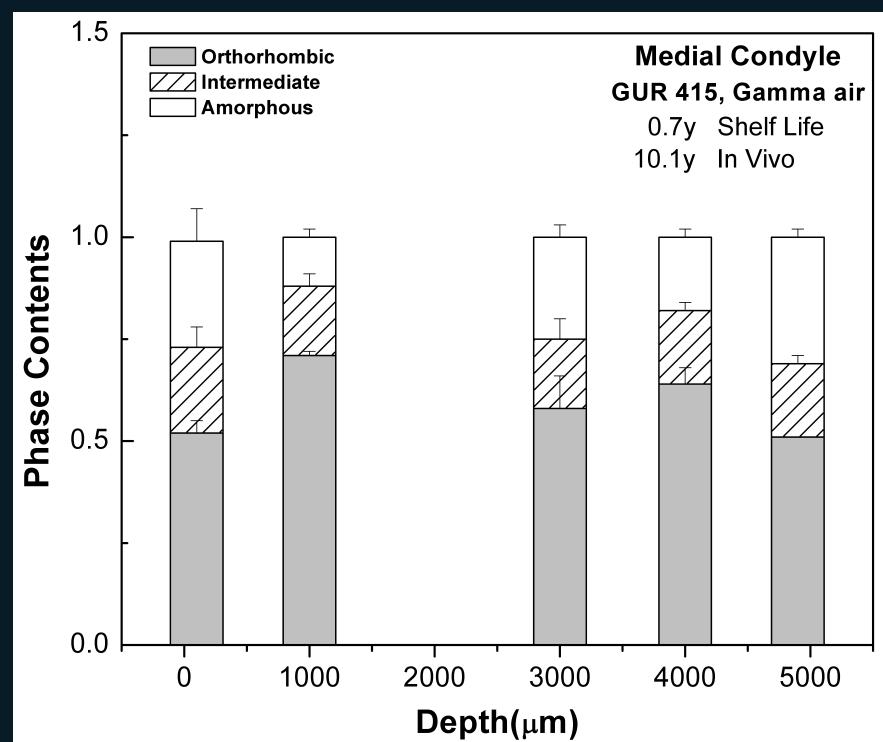
\* Paired t-tests

# RESULTS

## Raman spectroscopy

- Subsurface extrema (~ 1 mm)
  - ✓ Orthorhombic crystallinity ( $\alpha_c$ )
  - ✓ Overall crystallinity ( $\alpha_d$ )
  - ✓ Amorphous content ( $\alpha_a$ )
  - ✓ Intermediate fraction ( $\alpha_b$ )  
(Anomalous)
- Phase fractions higher/lower at the superior surface ( $p \leq 0.04^*$ )
- Higher %C and  $\alpha_c$  in the antero-posterior faces ( $p < 0.03^*$ )
- $\alpha_c$  and  $\alpha_d$  higher than %C ( $\dagger$ )

## THREE PHASE MODEL



\*Paired t-tests

†Student t-test

# RESULTS

## Crystallinity contents

Region \ Property	% C	$\alpha_c$	$\alpha_d$
<b>M. Anterior</b>	$0.64 \pm 0.17^*$	$0.68 \pm 0.16$	$0.73 \pm 0.07$
<b>M. Condyle</b>	$0.55 \pm 0.10$	$0.74 \pm 0.14^\dagger$	$0.71 \pm 0.07^\dagger$
<b>M. Posterior</b>	$0.70 \pm 0.20^*$	$0.79 \pm 0.07^*$	$0.74 \pm 0.06$
<b>C. Anterior</b>	$0.62 \pm 0.16$	$0.68 \pm 0.16$	$0.73 \pm 0.08$
<b>Ridge</b>	$0.55 \pm 0.09$	$0.67 \pm 0.14^\dagger$	$0.71 \pm 0.04^\dagger$
<b>Post</b>	$0.53 \pm 0.14$	$0.53 \pm 0.08$	$0.71 \pm 0.01$
<b>C. Posterior</b>	$0.59 \pm 0.11$	$0.69 \pm 0.09^\dagger$	$0.73 \pm 0.04^\dagger$

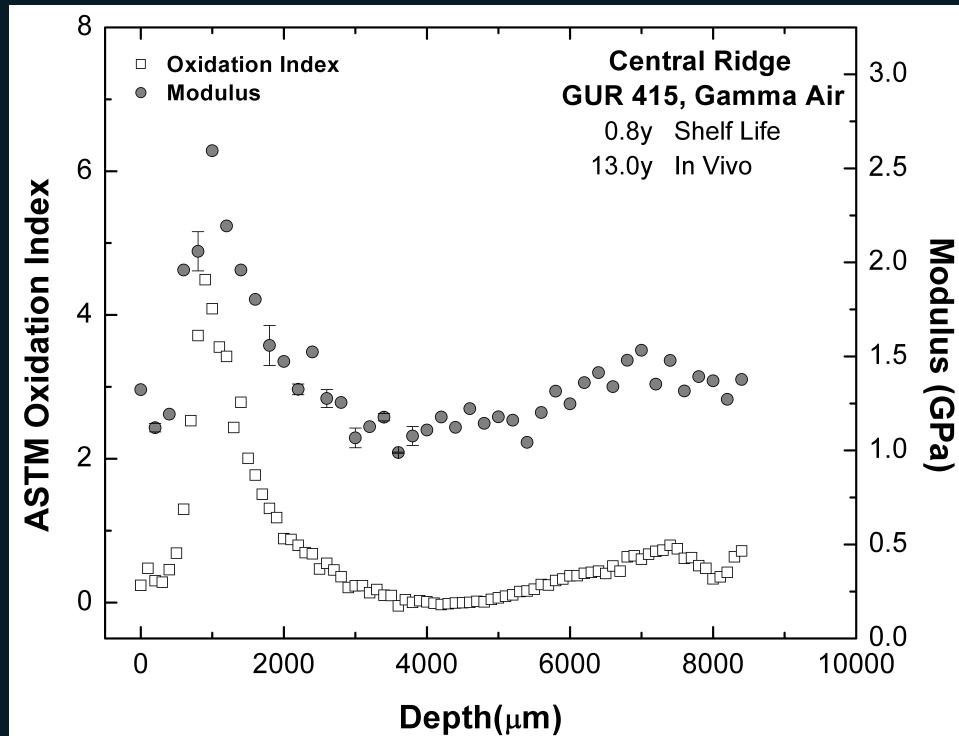
Antero-posterior faces: **Higher %C and  $\alpha_c$**  (Paired t-tests;  $p < 0.03^*$ )

$\alpha_c$  and  $\alpha_d$  higher than %C (Student t-test<sup>†</sup>)

# RESULTS

## Hardness and Modulus

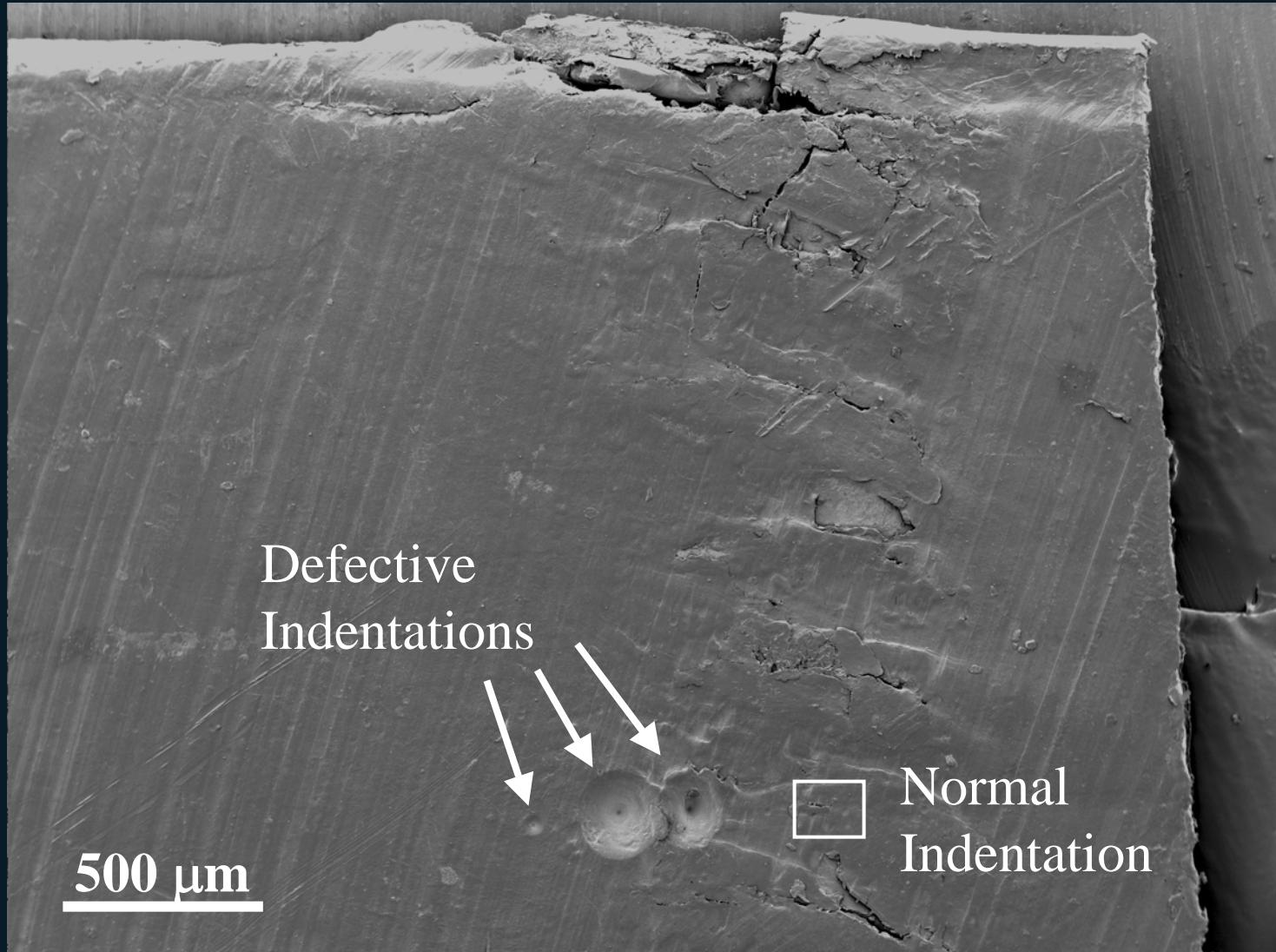
- Harder and stiffer material: **subsurface maxima** (~ 1 mm)
- No significant differences between superior and inferior surfaces ( $p = 0.3^*$ )
- Lower hardness and modulus at the **medial condyle** ( $p \leq 0.02^*$ )



\* Paired t-tests

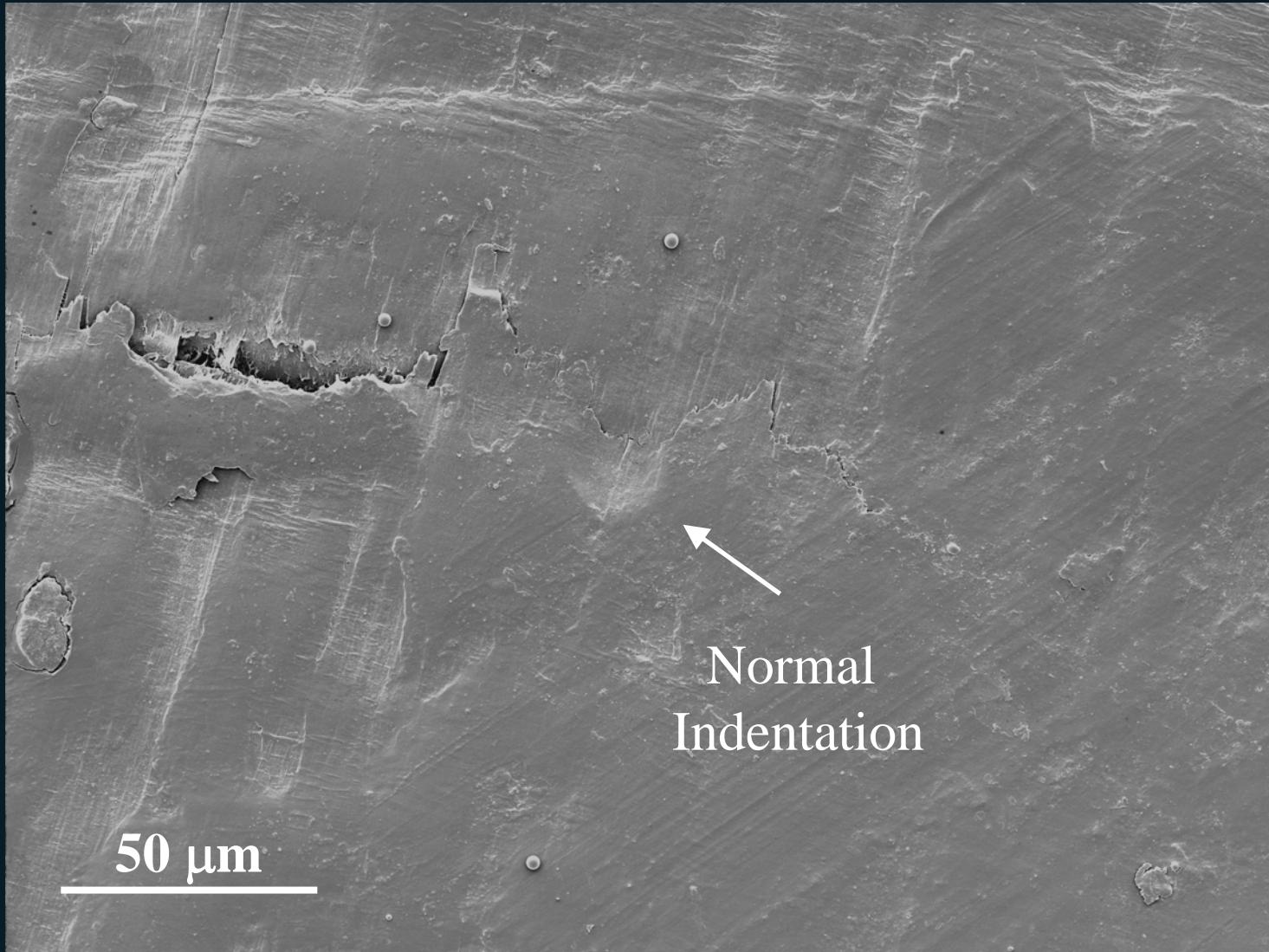
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## Nanoindentation



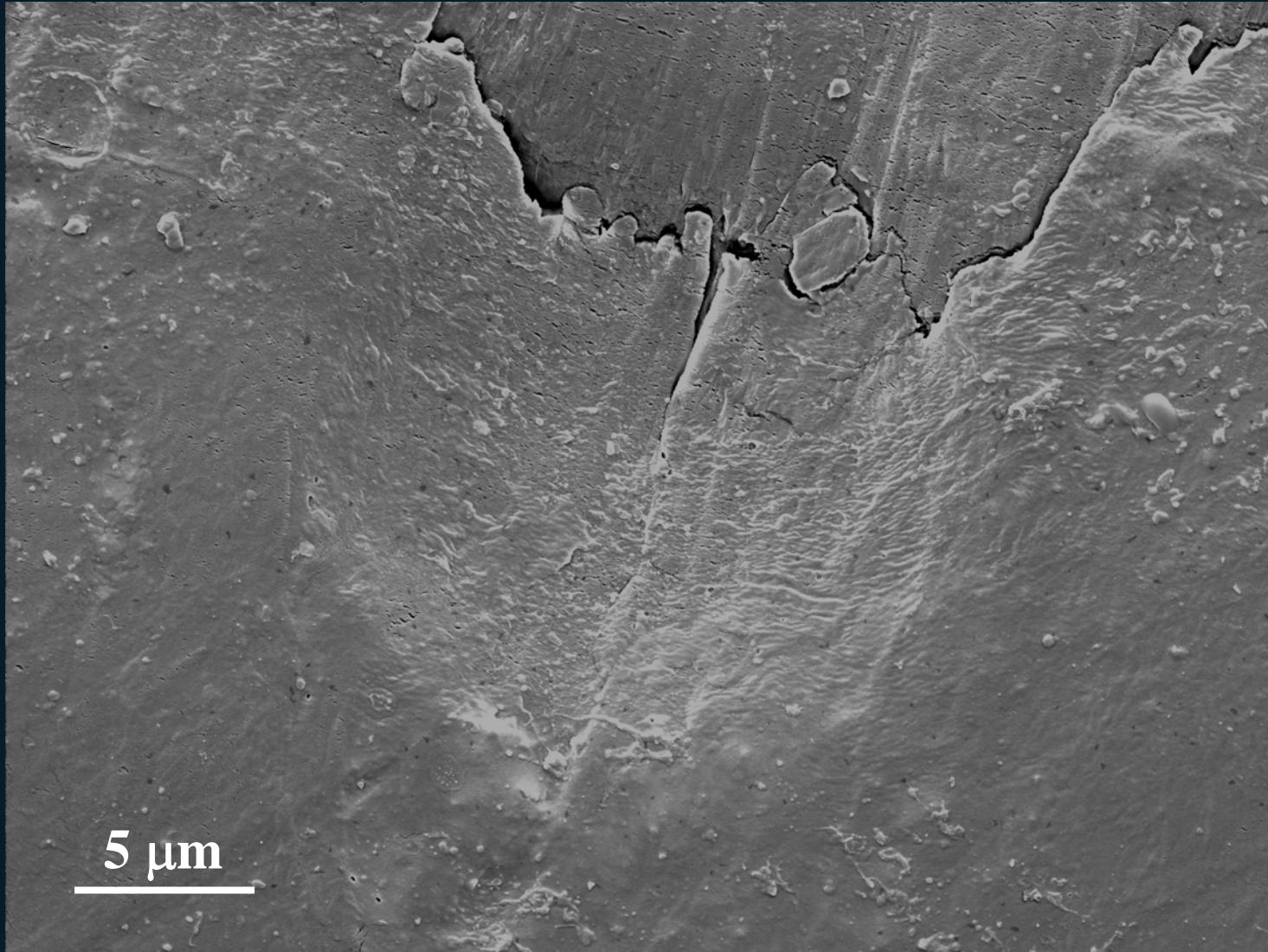
# RESULTS

## Nanoindentation



# RESULTS

## Nanoindentation



# DISCUSSION

- Historical TKA tibial inserts undergo oxidative, microstructural and mechanical degradation *in vivo*: access to body fluids is a key mechanism.
- FTIR and Raman spectroscopies confirm *in vivo* oxidation induces crystallinity changes
  - Qualitative estimations (dependence with density)
  - 3-Phase model not valid for highly oxidized PE
- Nanoindentation confirms evolution to a harder and stiffer PE, in spite of the high sensitivity to surface defects.

# **ACKNOWLEDGEMENTS**

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- Assistance of Zhorro Nikolov (Director at Drexel University Materials Characterization Facility) and Sandip Basu is highly appreciated.